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# **Optical absorption characteristics and polarization dependence of single-layer graphene on silicon waveguide**

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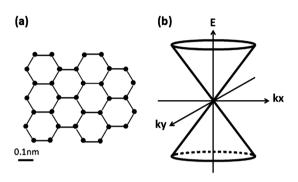
**SUMMARY** Graphene is attracting attention in electrical and optical research fields recently. We measured the optical absorption characteristics and polarization dependence of single-layer graphene (SLG) on submicrometer Si waveguide. The results for graphene lengths ranging from 2.5 to 200  $\mu$ m reveal that the optical absorption by graphene is 0.09 dB/ $\mu$ m with the TE mode and 0.05 dB/ $\mu$ m with the TM mode. The absorption in the TE mode is 1.8 times higher than that in the TM mode. An optical spectrum, theoretical analysis and Raman spectrum indicate that surface-plasmon polaritons in graphene support TM mode light propagation. *key words:* graphene, silicon photonics, polarizer, optical absorption

#### 1. Introduction

Graphene, a two-dimentional material with one-atom thickness [Fig. 1 (a)], is being investigated in electrical and optical research fields. Carbon atoms are strongly connected within each layer by covalent bonds of sp<sup>2</sup> orbitals, while they are connected weakly between layer with Van der Waals force. The interatomic distance and thickness are known to be 0.142 and 0.34 nm, respectively. Graphene's bandstructure has a unique shape as shown in Fig.1 (b), and graphene has various excellent characteristics [1]. In graphene, electrons are massless because of the linear dispersion relationship between energy and momentum at around the Dirac point. As a result, graphene has very high mobility (over 10,000 cm<sup>2</sup>/Vs) and is thus attracting attention in the electrical research field. Moreover, graphene's optical absorption is wavelength independent due to its zero band-gap structure, which enables us to fabricate broadband optical devices. This is the major reason it has recently attracted interest in the optical research field. Other excellent characteristics of graphene are a controllable band-gap in bi-layer graphene, high optical absorption efficiency ( $\pi \alpha \approx$ 2.3% per layer), various nonlinear optical effects (for example, four-wave mixing and saturable absorption), and unique

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**Fig.1** (a) Molecular structure of graphene. (b)Band structure of graphene.

light propagation modes. These characteristics of graphene have been exploited in the development of various electrical and optical graphene devices, such as high-speed transistors [2], THz light sources [3], fast photodetectors [4], high-efficiency modulators [5], and polarizers [6].

Recent advances in Si photonics enable us to implement a hybrid integrated wavelength division multiplexing (WDM) receiver as shown in Fig. 2(a) [7]. The receiver comprises a Si waveguides, SiOx arrayed waveguide grating, Ge photodetector, and electrical circuit on a  $1 \times 1$  cm<sup>2</sup> chip. It receives signal at 1.25 Gbit/s with 16 channels. In addition, monolithically integrated transceivers with Si photonics technology will become widespread widely in the future [Fig. 2(b)]. In the transceivers, electrical and optical circuits are integrated on the same substrate, so a CMOScompatible process is available, which strongly contributes to reducing fabrication costs. Since optical communications technologies are progressing rapidly, faster, more efficient, and smaller devices are required. However, previously reported Si modulators based on the carrier-plasmadispersion-effect have high power consumption and a large device area.

We assume that graphene can meet our expectations because it has many excellent characteristics as mentioned above, which enables the production of small-size devices with low power consumption. Small high-speed monolithically integrated transceivers with Si photonics technology and graphene [Fig. 2 (c)] will be implemented within 30 years [8].

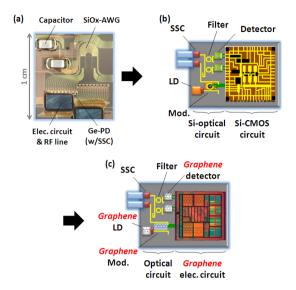
Table 1 compares various Si and graphene modulators [9]–[14]. In Table 1, case 1 and 2 indicate 10 Gbit/s on-chip optical channels with  $10^3$  channels and 50 Gbit/s with  $10^6$ 

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**Fig.2** Future vision of integrated devices. (a) WDM receiver with Si photonics. (b) Monolithically integrated transceiver with Si photonics. (c) Monolithic integrated transceiver with Si photonics and graphene.

**Table 1**Device characteristics comparisons of various Si modulatorsand graphene modulator. Case 1 is on-chip optical channels whose speedand number of channels are 10 Gbit/s and  $10^3$  channels. Case 2 is on-chipoptical processing whose speed and number of channels are 50 Gbit/s and  $10^6$  channels.

		<b>Si —</b> [9],				Graphene [13], [14]
	Туре	Forward	Reverse	Forward	Reverse	-
	Area (µm²)		<b>1 x 10</b> <sup>4</sup>	1 x 10²	1 x 10 <sup>3</sup>	3 x 10
	Power (fJ/ bit)	5 x 10 <sup>3</sup>	3 x 104	0 <sup>4</sup> 3 x 10 <sup>2</sup> 5 x 10		7 x 10 <sup>-1</sup>
Ва	Bandwidth (GHz)		20	25	11	1
Case 1	Area/ chip (μm²)	1	1 x 10	1 x 10 <sup>-1</sup>	1	1 x 10 <sup>-2</sup>
Case I	Power/ chip (W)	5 x 10	3 x 10 <sup>2</sup>	3	5 x 10 <sup>-1</sup>	7 x 10 <sup>-3</sup>
Case 2	Area/ chip (μm²)	1 x 10 <sup>3</sup>	<b>1 x 10</b> <sup>4</sup>	L x 10 <sup>4</sup> 1 x 10 <sup>2</sup> 1 x 10	1 x 10 <sup>3</sup>	3 x 10
Case 2	Power/ chip (W)	2.5 x 10⁵	1.5 x 10 <sup>6</sup>	1.5 x 104	2.5 x 10 <sup>3</sup>	3.5 x 10

channels, respectively. The former can be implemented with Si photonics technology, but the latter is problematic and difficult to realize because of its huge power consumption and large area. Even if the most efficient modulation scheme with a reverse-biased Si-ring modulator is used, the power consumption reaches 60 kWh, which is three times what one person uses in a day in Japan [15], [16]. This problem could be mitigated by using graphene-based modulators, whose power consumption and area per chip are smaller than those with Si photonics technology only.

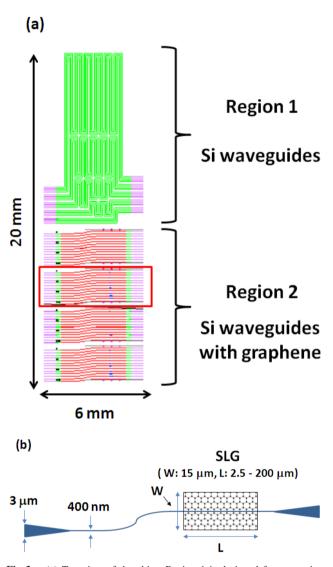
There are two reasons why a graphene modulator on a Si platform provides high efficiency. One is that a graphene modulator enables a high extinction ratio and wide bandwidth due to its high optical absorption efficiency per layer and high mobility compared with conventional materials. Moreover, the Fermi level and optical absorption coefficient can be controlled by applying a low swing voltage. The other reason is that a Si waveguide used as an optical platform strongly contributes to increasing the overlap between graphene and Si core because the intensity of evanescent light of the Si waveguide is stronger than that of other waveguides, which enables light to interact with graphene more strongly. As a result, an ultra-short (several ten micrometers) and high-efficiency device can be realized. By integrating graphene and Si devices, we can achieve small, high-speed and low-power consumption integrated devices with higher order channels.

However, the detailed characteristics of graphene on Si waveguides have not been investigated. In the Ref. [13], a graphene modulator on a Si waveguide was measured with only TE mode light; the optical absorption characteristics and polarization dependence of graphene were not characterized. Recently, we have transferred single-layer graphene with few defects onto Si waveguides and characterized its optical absorption efficiency in TE and TM mode polarizations [17]. In this report, we describe additional information, including scanning electron microscopy (SEM) of single-layer graphene on a Si waveguide, propagation and coupling losses of Si waveguide and theoretical calculation of graphene's conductivity to confirm mechanism of light propagation in graphene.

In this report, first, we will describe about device layout and fabrication in section 2. Next, we will show the results of micro-Raman spectroscopy and optical measurement in section 3 and 4. Finally, we will give some discussion in section 5 and conclude our research in section 6.

#### 2. Device Layout and Fabrication

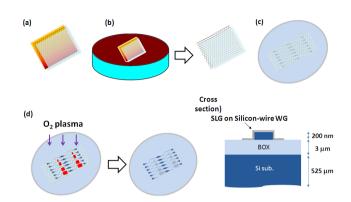
Before describing our device layout and fabrication process, we review the three main methods of graphene growth. Graphene can be obtained by exfoliation from highly oriented pyrolytic graphite (HOPG) [18], by epitaxial growth on SiC substrate [19], and by chemical vapor deposition (CVD) on a transition metal such as Cu and Ni, which uses a catalyzed process [20]. Graphene obtained by exfoliating HOPG is smaller than 1,000  $\mu$ m<sup>2</sup> and difficult to transfer to arbitrary positions, though it is high quality, nondoping and has high mobility (~10,000 cm<sup>2</sup>/Vs). However, to fabricate graphene-based optical devices, large-scale graphene is needed. To make multi-layer graphene, epitaxial growth on a SiC substrate is used. Large-scale graphene can be obtained by this method, but the mobility is low (~1,100 cm<sup>2</sup>/Vs) and there are relatively numerous defects. Moreover, it is difficult to transfer graphene grown on SiC substrate to other substrates because of strong interaction between graphene and its substrate. Compared to graphene grown by these two methods, graphene grown by CVD is large scale, easy to transfer onto arbitrary substrates, and has few defects. In addition, the mobility is higher (up to 4,050  $cm^2/Vs$ ) than that of graphene grown by the epitaxy method, and it is easy to control the number of layers. For these reasons, CVD graphene is appropriate for use as a device material. If we use a Cu substrate, the low solubility of carbon



**Fig.3** (a) Top view of the chip. Region 1 is designed for measuring the propagation and coupling losses in Si waveguides, which have different lengths. Region 2 is designed for measuring the absorption in the partly integrated graphene ranging from 2.5 to  $200 \,\mu\text{m}$ . (b) Top view of the graphene device.

in Cu helps self limiting in the growth process and singlelayer graphene can be synthesized. On the other hand, fewlayer graphene can be synthesized on Ni due to the high solubility of carbon. Direct graphene growth on Si substrate has also been investigated recently because of its simplicity, but single-layer graphene synthesis has not been achieved yet [21]. Therefore, CVD-grown graphene on Cu is suitable for the fabrication of optical devices based on singlelayer graphene, and we selected CVD-grown graphene for our fabrication process.

Our device layout on a 20 mm  $\times$  6 mm chip is shown in Fig. 3 (a). Region 1 is Si waveguides without graphene integration, which is used to measure the propagation and coupling losses by varying the length of the waveguides. Region 2 is Si waveguides with graphene integration, and an expanded view of the area enclosed by the square is



**Fig. 4** Fabrication process. (a) PMMA was spincoated on a CVD-grown graphene on Cu substrate. (b) Cu substrate was etched with FeCl<sub>3</sub> solution. (c) Graphene was transferred onto all Si waveguides on a 4-inch SOI wafer. (d) PMMA was etched and graphene pattern was defined by photolithography and RIE. Cross section) Schematic of cross section of graphene device after all fabrication process.

shown in Fig. 3 (b). The line is a Si waveguide whose length, width and height are 6 mm, 400 nm, and 200 nm, respectively. We selected this width and height because they satisfy the single-mode propagation condition and yield evanescent light which has strong intensity at TE and TM polarizations. Tapered waveguides are formed at the input and output to reduce coupling losses. In this research, we were unable to form inverse taper spot size converters (SSCs) because the overclad process with electro cyclotron resonance-plasma chemical vapor deposition (ECR-CVD) damages graphene. The length of the single-layer graphene on Si waveguide is varied as 0, 2.5, 10, 20, 30, 50, 80, and 200  $\mu$ m and width is fixed at 15  $\mu$ m.

To fabricate the graphene device, we started from a 4-inch silicon on insulator (SOI) wafer (thickness of Si substrate, SiO<sub>2</sub>, and Si core are 525  $\mu$ m, 3  $\mu$ m and 200 nm, respectively). After a resist masked pattern had been formed by electron-beam (EB) lithography, Si channel waveguides were precisely defined by electro cyclotron resonance-plasma ion etching (RIE) [22]. The fabrication process with graphene is shown in Fig. 4. First, we prepared  $80 \times 80$ -mm<sup>2</sup> thermal CVD grown single-layer graphene on a Cu substrate and cut it into pieces  $30 \times 30$  mm<sup>2</sup>. Next, we spincoated PMMA film with thickness of several hundred nanometers to support the graphene [Fig. 4 (a)]. After that, we baked it at 170°C and etched the Cu substrate in ferric chloride (FeCl<sub>3</sub>) solution for over 12 hours at room temperature [Fig. 4 (b)]. Next, we rinsed the samples several times and transferred two sheets of graphene next to each other in deionized water to cover all Si waveguides on the wafer [Fig. 4 (c)]. Then, we dissolved the PMMA with acetone. Finally, we formed a  $1.5-\mu$ m-thick photoresist and used photolithography and RIE with O<sub>2</sub> gas to define the graphene patterns [Fig. 4 (d)].

### 3. Micro-Raman Spectroscopy

After transferring the graphene, we confirmed the quality and pattern accuracy of graphene by SEM observation and

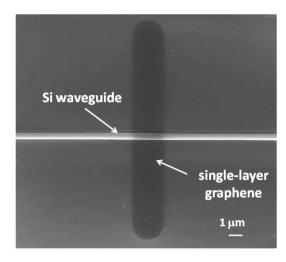
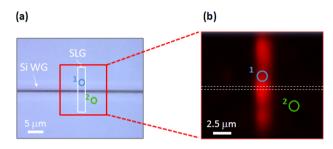


Fig. 5 SEM image of graphene on Si waveguide.



**Fig.6** Microscope image and micro-Raman spectroscopy (a) Microscope image of transferred single-layer graphene whose length is 2.5  $\mu$ m on Si waveguide. (b) Raman mapping of 2D band.

micro-Raman spectroscopy. An SEM image of  $2.5 - \mu$ m-long transferred graphene is shown in Fig. 5. We can observe that the transferred graphene has few cracks and the pattern is formed accurately. Then, we used 532-nm light as a pump and an objective lens of  $50 \times to$  observe the transferred graphene by micro-Raman spectroscopy. Figure 6 (a) shows an optical microscope image of 2.5-µm-long singlelayer graphene on a Si waveguide. We selected the smallest graphene length to confirm that our fabrication process is sufficient for defining narrow graphene patterns. Figure 6 (b) shows an expanded view of the area enclosed by the square together with Raman mapping of the 2D band  $(2695 \text{ cm}^{-1})$ . The bright area indicates the position of the graphene, which accurately reflects the photomask patterns as designed. Figure 7 shows the Raman-shift spectrum for measurement points 1 and 2 in Fig. 6. At point 1, the Raman characteristics of graphene appeared [23]. The D band (1350 cm<sup>-1</sup>), which represents the existence of defects almost does not appear. In addition, information about the number of graphene layers can be attained from the peak intensity ratio of the G (1592 cm<sup>-1</sup>) and 2D bands. In this case, I(G)/I(2D) was approximately 0.25, so the transferred graphene is single layer. At point 2, the spectrum indicates that transferred graphene was perfectly etched in the un-masked region. From these results, we confirmed that transferred graphene is single layer with few defects and the graphene pattern is formed precisely. Additionaly, we also

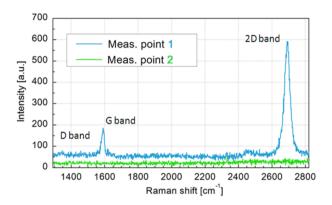


Fig. 7 Raman-shift spectrum for measurement points 1 and 2.

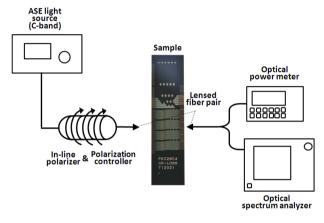


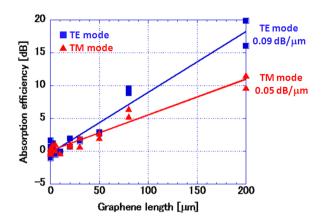
Fig. 8 Optical measurement setup.

confirmed that the coverage of the single-layer graphene is over 90%, with hardly any multiple-layers or cracking.

#### 4. Optical Measurement

We measured the optical absorption characteristics and polarization dependence of single-layer graphene on a Si waveguide. The optical measurement setup is shown in Fig. 8. We used an amplified spontaneous emission (ASE) light source in the C-band (1530-1565 nm) and lensed fibers. Because the ASE light source has a large bandwidth, Fresnel reflection due to resonance can be prevented. A high-input intensity may cause saturable absorption in graphene and two-photon-absorption induced free-carrier absorption in Si waveguide, so we set the on-chip coupled power to approximately 3 mW for the TE mode and 0.6 mW for the TM mode to avoid undesired nonlinear optical effects. After polarization control with a polarizer and polarization controller, the input light was coupled into the device. Optical absorption and spectrum of sample were measured with a power meter and optical spectrum analyzer.

Figure 9 shows the results of a measurement. Optical absorptions by graphene in TE and TM modes are 0.09 and 0.05 dB/ $\mu$ m, which were evaluated from the slopes. We obtained a high extinction ratio per unit length of 0.04 dB/ $\mu$ m, which is the difference in optical absorption between the TE and TM modes.



**Fig.9** Measurement results for optical absorption and polarization dependence of graphene (The coupling loss is approximately 7 dB with the TE mode and 15 dB with the TM mode.).

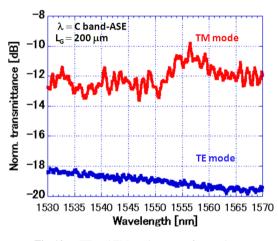
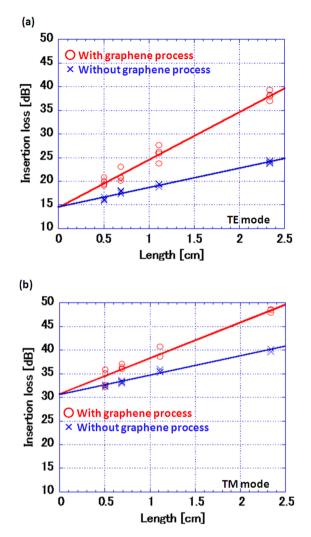


Fig. 10 TE and TM mode spectra for graphene.

TE and TM mode spectra of  $200-\mu$ m-long graphene are shown in Fig. 10. The spectra were obtained from the difference between the reference spectrum and that of the graphene-integrated Si waveguide at the same device length. In the TM mode, a small decrease in absorption (1~2 dB) around 1550 to 1560 nm was observed, which can be explained by slight nonlinear absorption in the reference Si waveguide. The spectrum range actually agrees well with the gain peak of the ASE light source, as shown in Ref. [17]. In addition, we measured the propagation and coupling losses of Si waveguides with and without the graphene process, as shown in Fig. 11 and Table 2. The propagation loss with the graphene process is twice as large as that without it. We suppose the cause of this is that a few graphene particles remain on the Si waveguide, which originate from the few-layer part of the transferred graphene. Note that the sidewall and top-surface roughness of the Si waveguide would be limited, because the etching time was quite short (only about several minutes). Moreover, the coupling loss for each port is very large, approximately 7 dB/facet with the TE mode and 15 dB/facet with the TM mode, and these values are the same with and without graphene processes. We have to improve our transfer processes to prevent graphene



**Fig. 11** (a) Loss of Si waveguide with and without graphene process (TE mode). (b) Loss of Si waveguide with and without graphene process (TM mode).

particles and design inverse tapers to reduce coupling losses [22].

#### 5. Discussion

Since graphene interacts with propagating electric field, we simulated the mode profiles in a Si waveguide as shown in Fig. 12. The electric field of evanescent light in the TM mode is stronger than that in TE mode because of the flatterned waveguide. We anticipated that optical absorption for the TM mode would be larger than that for the TE mode as a result of the very strong interaction between light and graphene. However, the results of the measurement were the opposite. Here, we focused on the spectra of the two polarization modes in graphene (Fig. 10). In the TE mode, absorption induced by graphene increases because the mode becomes larger as wavelength increases; the evidence indicates that TE mode light is simply absorbed by the graphene. However, the absorption of TM mode light is wavelength independent, which is unusual. Here, we discuss the mech-

 Table 2
 Comparison of propagation and coupling losses with and without graphene process.

		With gr. process	Without gr. process
TE mode	Propagation loss [dB/cm]	10.1	4.1
TE mode	Coupling loss [dB/facet]	7.2	7.2
TM mode	Propagation Loss [dB/cm]	7.6	4.1
TM mode	Coupling Loss [dB/facet]	15.3	15.3

anism of TM mode absorption on the basis of a theoretical calculation.

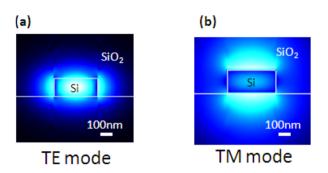
Photoconductivity of graphene is represented by the sum of  $\sigma_{inter}$  and  $\sigma_{intra}$  which are represented by

$$\sigma_{\text{inter}}(\omega) = \frac{je^2}{4\pi\hbar} \ln \left( \frac{2\,|\mu| - \left(\omega + j\tau^{-1}\right)\hbar}{2\,|\mu| + \left(\omega + j\tau^{-1}\right)\hbar} \right) \tag{1}$$

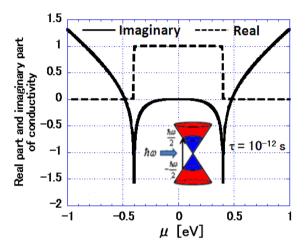
$$\sigma_{\text{intra}}(\omega) = \frac{je^2\mu}{\pi\hbar^2(\omega + j\tau^{-1})}$$
(2)

where e is electric charge,  $\hbar = h/2\pi$  is the reduced Planck's constant,  $\mu$  is the Fermi level ( $\neq 0$ ),  $\omega$  is the radian frequency of input light whose wavelength is 1.55  $\mu$ m, and  $\tau$  is the relaxation time of graphene [6]. If the imaginary part of  $\sigma$  is negative, optical absorption of TM mode is large and propagation is mainly in the TE mode; if it is positive, the TM mode propagates mainly. We calculated the imaginary part of conductivity on Fermi level with Eqs. (1) and (2) as shown in Fig. 13 (We calculated with  $\tau = 10^{-12}$  s) [24]–[26]. When the Fermi level is within  $\pm 0.48$  eV, the propagation is in the TE mode because the imaginary part of conductivity is negative. In band structure of graphene, the area is called an interband ( $|\mu| \le \hbar \omega/2$ ). The interband transition occurs and the TE mode propagates. However, when the Fermi level is not around -0.4 or +0.4 eV, propagation is not a complete TE mode because the imaginary part of conductivity is approximately zero. In the other range ( $|\mu| > 0.48$  eV), propagation is in the TM mode because the imaginary part of the conductivity is positive and the graphene acts as metal. In the band structure of graphene, there is an area known as the intraband  $(|\mu| > \hbar \omega/2)$ . In the intraband, free carriers and electric field interact and the TM mode propagates by means of surface plasmon polaritons, which strongly contribute to the low absorption in TM propagation compared with that in the TE mode [27].

Then, we estimated the Fermi level from the micro-Raman spectrum. As described in Ref. [28], the peak shifts of G and 2D bands occur as a result for an effect of carrier doping in graphene. Comparing with Fig. 7 and Ref. [28], we confirmed that the transferred graphene exhibits pdoping (Fermi level is approximately -0.4 - 0.5 eV). It is supposed that the graphene acts as a metal and the propagation is in the TM mode due to surface Plasmon polaritons.



**Fig. 12** Simulation of electric field in Si waveguide without graphene integration. (a) TE mode. (b) TM mode.



**Fig.13** Imaginary part and real part of conductivity of graphene in units of  $e^2/4\hbar$ . A solid line and dotted line indicates the imaginary and real part of  $\sigma$ . Inset: Band structure of graphene.

It was reported that graphene on Si substrate becomes p-doped when it touches water because OH radicals bond with Si and promote carrier exchange [29]. The graphene would become p-doped during its transfer to a wafer.

Finally, we discuss the device's working as a polarizer. We obtained a four times higher extinction ratio (0.04 dB/ $\mu$ m) than that in Ref. [6], in which the graphene was integrated on a standard single-mode fiber (~10.6 dB/mm). The advantage is that intensity of evanescent light of the Si waveguide is stronger. To obtain a 20-dB extinction ratio, millimeter-order graphene was required in Ref. [6], but in this work, we observed equivalent characteristics in graphene of several-hundred micrometers. The Fermi level of graphene exists on the boundary between TE and TM propagation in this work. We will be able to obtain a higher extinction ratio by changing Fermi level.

## 6. Conclusion

Single-layer graphene was successfully transferred to Si waveguides and the graphene pattern was formed precisely by RIE. We confirmed the optical absorption and polarization dependence of single-layer graphene on a Si waveguide. The results revealed that TE mode absorption is approximately 1.8 times higher than that of TM mode because sur-

face plasmon polaritons support the TM mode. We suppose that if we add a gate oxidation layer and electrodes to the device, the polarization can be changed by applying gate voltage because the Fermi level of graphene changes. We expect to be able to fabricate a high-efficiency, controllable and ultra-small polarizer. The fundamental optical properties of graphene, clarified in this report will contribute to the future development of graphene-integrated photonic de-

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